

# The analysis of Au/TiO<sub>2</sub>/n-Si Schottky barrier diode at high temperatures using I-V characteristics

B. KINACI\*, T. ASAR, Y. ÖZEN, S. ÖZÇELİK

Department of Physics, Gazi University, 06500, Ankara, Turkey

In this study, the current-voltage (*I-V*) characteristics of Au/TiO<sub>2</sub>/n-Si Schottky barrier diodes (SBDs) were examined at high temperatures. TiO<sub>2</sub> thin films were deposited on polycrystalline n-type Silicon (Si) substrate using DC magnetron sputtering system. In order to improve the crystal quality, thermal annealing process were done at 700 °C. The electrical parameters such as barrier height ( $\Phi_b$ ), ideality factor (*n*) and series resistance ( $R_s$ ) of Au/TiO<sub>2</sub>/n-Si SBDs have been investigated by using forward and reverse bias *I-V* measurements in the temperature range of 340-400 K by steps of 20 K. Also, the values of  $R_s$  and  $\Phi_b$  were determined by using Cheung's and Norde methods. It was seen that there was a good agreement between the values of  $R_s$  and  $\Phi_b$  obtained from the forward bias  $\ln(I-V)$  curves by applying Cheung's and Norde methods.

(Received March 17, 2011; accepted April 11, 2011)

**Keywords:** Schottky barrier diode, TiO<sub>2</sub>, Temperature dependence, DC magnetron sputtering

## 1. Introduction

TiO<sub>2</sub> is a wide band gap semiconductor material which has attracted considerable attention for its potential applications in optical components such as photocatalysis, gas sensor, solar cell and optical fiber [1-4]. It has three types of bulk crystal structure in TiO<sub>2</sub>: anatase, rutile and brookite [5]. Among these crystal, anatase TiO<sub>2</sub> has attracted a great deal of interest because of its excellent photocatalytic behaviour. It is known that the deposition method and calcination temperatures determine the physical properties of the TiO<sub>2</sub> material. Depending on the calcination temperature TiO<sub>2</sub> films transform first from amorphous to anatase and then rutile. [6]. Although a variety of methods have been used to prepare TiO<sub>2</sub> films, sputtering is one of the most utilized method. [7].

Recently, there were published many studies on SBDs which has an important role in modern device technology [8-16]. The electrical properties of the SBDs depend especially on the series resistance of device. There are various techniques to calculate the main electrical parameters such as forward bias *I-V* measurements, Cheung's method [17] and Norde method [18]. Electronic properties of a Schottky diode are characterized by its main electrical parameters such as *n*,  $\Phi_b$  and  $R_s$  which are parameters give useful information concerned with the nature of the diode. Analysis of the *I-V* characteristics of SBDs based on thermionic emission theory usually reveals an abnormal decrease in the BH and an increase in the ideality factor with a decrease in temperature [19, 20].

In this study, *I-V* characteristics of Au/TiO<sub>2</sub>/n-Si SBDs with anatase phase TiO<sub>2</sub> were measured in the temperature range of 340-400 K by steps of 20 K. The temperature dependence of electrical parameters such as *n*,  $\Phi_b$  and  $R_s$  were extracted from forward bias *I-V* measurement. Also,  $\Phi_b$  and  $R_s$  were calculated using Cheung's and Norde methods. It was seen that there was a

good agreement between the values of  $R_s$  and  $\Phi_b$  obtained from the forward bias  $\ln I-V$  curves by applying Cheung's and Norde methods.

## 2. Experimental procedure

Au/TiO<sub>2</sub>/n-Si SBDs were fabricated on a 2" diameter with (100) orientation, 350 μm thickness, 0.01 Ω.cm resistivity and phosphorus doped (n-type) polycrystalline Si substrate. Si substrate was

degreased using organic solvent. After then, the Si substrate was mounted onto the stainless steel optically heated sputtering holder and loaded into the DC magnetron sputtering system. After the preparation of the substrate to the deposition of 1500 Å thickness TiO<sub>2</sub> thin film, the Si substrate was transferred into the deposition chamber. TiO<sub>2</sub> thin film was deposited using high purity (99.999%) Ti target, under specific Ar+O<sub>2</sub> reactive gas mixture (Ar/O<sub>2</sub>=90/10 sccm) controlled with mass flow controllers. For the deposition, the substrate temperature and the pressure was set to 200 °C and 4.2×10<sup>-3</sup> mbar, respectively and kept constant during the whole deposition. The thermal annealing of the sample was carried out in air for 4 h at 700 °C. For the electrical characterization, in the thermal evaporation system ohmic and rectifier contacts were formed using high purity Au (99.999%).

The forward and reverse bias *I-V* measurements were performed in the temperature range of 340 - 400 K using a Keithley 2400 source-meter. The temperature was adjusted using Janis vpf-475 cryostat and a Lake Shore model 321 auto-tuning temperature controllers with sensitivity better than ± 0.1 K. The whole electrical measurements were also performed using micro computer through an IEEE-488 AC/DC converter card.

### 3. Results and discussion

The current-voltage ( $I$ - $V$ ) characteristics of Au/TiO<sub>2</sub>/n-Si SBDs in the temperature range of 340-400 K by steps of 20 K are given in Fig. 1. In order to quantitatively analyze the Au/TiO<sub>2</sub>/n-Si SBDs we assume the standard thermionic emission (TE) theory as follows [21]

$$I = I_o \exp\left(\frac{q(V - IR_s)}{nkT}\right) \quad (1)$$

Where the  $IR_s$  term is the voltage drop across the  $R_s$  of structure,  $V$  is the applied voltage across to rectifier contact,  $I_o$  saturation current is defined by

$$I_o = AA^*T^2 \exp\left(-\frac{q\Phi_b}{kT}\right) \quad (2)$$

The quantities  $A$ ,  $A^*$ ,  $T$ ,  $k$  and  $q$  are the diode area, the effective Richardson constant of 112 A/cm<sup>2</sup>K<sup>2</sup> for n-type Si, temperature in Kelvin, Boltzmann constant and electronic charge, respectively.

Once  $I_o$  is determined the  $\Phi_b$  is obtained by rewriting Eq. (2) as

$$\Phi_b = \frac{kT}{q} \ln\left[\frac{AA^*T^2}{I_o}\right] \quad (3)$$

Table 1. Temperature dependent of various parameters determined from  $I$ - $V$  characteristics of Au/TiO<sub>2</sub>/n-Si SBDs.

T (K)	$\Phi_b$ (I-V) (eV)	$R_s$ (dV/dln(I)) ( $\Omega$ )	$\Phi_b$ (H(I)) (eV)	$R_s$ (H(I)) ( $\Omega$ )	$\Phi_b$ (F(V)) (eV)	$R_s$ (F(V)) ( $\Omega$ )
340	0.74	179.85	0.78	199.07	0.78	156.4
360	0.78	172.65	0.86	211.08	0.82	170.92
380	0.82	178.81	0.88	231.24	0.84	212.59
400	0.85	184.33	0.93	217.03	0.88	217.89

Using Eqs. (3) and (4) the values of  $\Phi_b$  and  $n$  of the Au/TiO<sub>2</sub>/n-Si SBDs at different temperatures were calculated. The experimental values of  $\Phi_b$  and  $n$  of Au/TiO<sub>2</sub>/n-Si SBDs were changed from 0.74 eV and 2.47 at 340 K to 0.85 eV and 2.24 at 400 K, respectively. Also,  $R_s$  values were determined from the  $I$ - $V$  characteristics.  $R_s$  values of Au/TiO<sub>2</sub>/n-Si SBDs were changed from and 288.24  $\Omega$  to 318.56  $\Omega$  at 340 K and 400 K, respectively. The  $\Phi_b$  calculated from forward bias  $I$ - $V$  characteristics shows an unusual behavior that increase with the increase of temperature. Similar results have been reported in the literature [15]. Such temperature dependence was an

The  $n$  can be calculated from the slope of the linear region of the forward bias  $\ln(I$ - $V)$  plot for each temperature and can be written from Eq.(1), as

$$n = \frac{q}{kT} \left( \frac{dV}{d(\ln I)} \right) \quad (4)$$

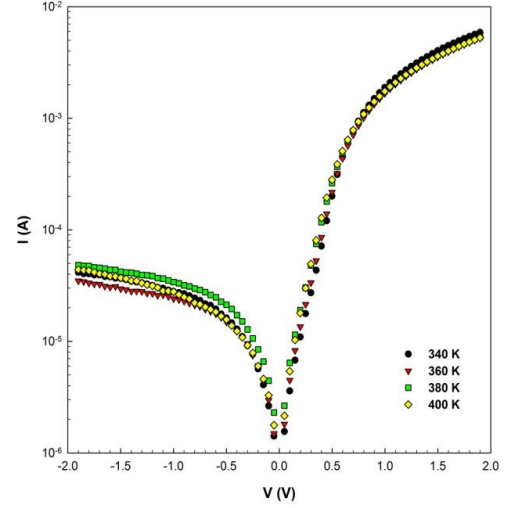


Fig. 1. Temperature dependent semi-logarithmic plot of the  $\ln I$ - $V$  characteristics of Au/TiO<sub>2</sub>/n-Si SBDs with anatase phase TiO<sub>2</sub> at various temperatures.

obvious disagreement with the reported negative temperature coefficient of the Schottky barrier height.

Series resistance ( $R_s$ ) and barrier height ( $\Phi_b$ ) were calculated from the forward bias  $I$ - $V$  data using methods developed by Cheung and Cheung [17]. Cheung's function

$$\frac{dV}{d\ln I} = n \frac{kT}{q} + R_s I \quad (5)$$

$$H(I) = V - n \frac{kT}{q} \ln \left( \frac{I_0}{AA^*T^2} \right) = n\Phi_b + R_s I \quad (6)$$

The experimental  $dV/d\ln(I)$  versus  $I$  and  $H(I)$  versus  $I$  plots of at different temperatures are given in Fig. 3 (a) and (b), respectively.

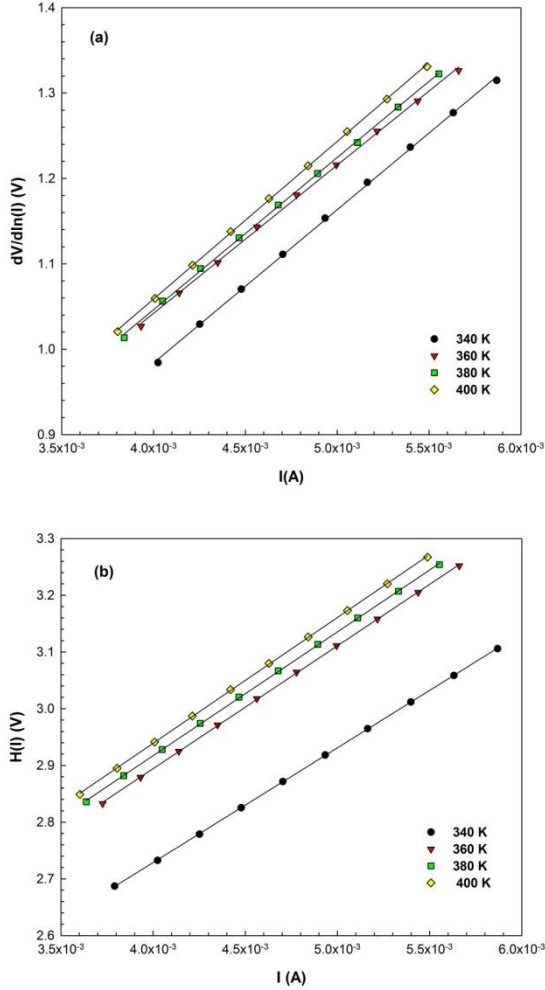


Fig. 2. (a) Experimental  $dV/d\ln(I)$  vs  $I$  and (b)  $H(I)$  vs  $I$  plots of Au/TiO<sub>2</sub>/n-Si SBDs at various temperatures.

The values of series resistance of Au/TiO<sub>2</sub>/n-Si SBDs were calculated from both  $dV/d\ln I$  versus  $I$  plot according to Eq. (5) and  $H(I)$  versus  $I$  plot according to Eq. (6) in the temperature range of 340 K to 400 K. The determined values of  $R_s$  and  $\Phi_b$  are given in Table 1. As can be seen, the determined values of  $R_s$  ( $dV/d\ln(I)$  versus  $I$ ) of Au/TiO<sub>2</sub>/n-Si SBDs were changed from 179.85  $\Omega$  to 184.33  $\Omega$  at 340 K and 400 K, respectively. The determined values of  $R_s$  and  $\Phi_b$  ( $H(I)$  versus  $I$ ) of Au/TiO<sub>2</sub>/n-Si SBDs were changed from 199,07  $\Omega$  and 0,78 eV at 340 K to 217,03  $\Omega$  and 0,93 eV at 400 K, respectively. It was seen in Table 1, The values of  $R_s$  obtained from  $dV/d\ln(I)$  versus  $I$  and  $H(I)$  versus  $I$  plots are in good agreement with each other. Also, the difference in

the values of  $\Phi_b$  obtained from Eqs (3) and (6) is an effect of existence of  $R_s$  and thickness interfacial layer

Norde proposed an alternative method to determine the value of the  $R_s$  and  $\Phi_b$ . The values of  $R_s$  and  $\Phi_b$  were obtained from the modified Norde method developed by Bohlin [22]. The modified Norde method is expressed as

$$F(V, \gamma) = \frac{V}{\gamma} - \frac{q}{kT} \ln \left( \frac{I}{AA^*T^2} \right) \quad (7)$$

where  $I$  is current obtained from the  $I$ - $V$  curve,  $\gamma$  is the dimensionless value greater than ideality factor. A plot of  $F(V)$  versus  $V$  for Au/TiO<sub>2</sub>/n-Si SBDs at different temperatures are given in Fig. 3 From the plot of  $F(V)$  versus  $V$ , the value of  $\Phi_b$  can be determined as follows

$$\Phi_b = F(V_0) + \frac{V_0}{\gamma} \quad (8)$$

where  $F(V_0)$  is the minimum point of  $F(V, \gamma)$  versus  $V$  and  $V_0$  is the corresponding bias voltage.  $R_s$  can be expressed as

$$R_s = \left( \frac{kT}{qI} \right) (\gamma - n) \quad (9)$$

From the  $F(V)$  versus  $V$  plots the values of  $R_s$  and  $\Phi_b$  were changed from 165,4  $\Omega$  and 0,78 eV at 340 K to 217,89  $\Omega$  and 0,88 eV at 400 K, respectively, by using the values  $F(V_0)$  and  $V_0$  were changed from 0,687 V and 0,599 V at 340 K to 0,798 V and 0,549 V, respectively. The  $R_s$  and  $\Phi_b$  values from the modified Norde and Cheung's methods have been compared with each other and results were given in Table 1.

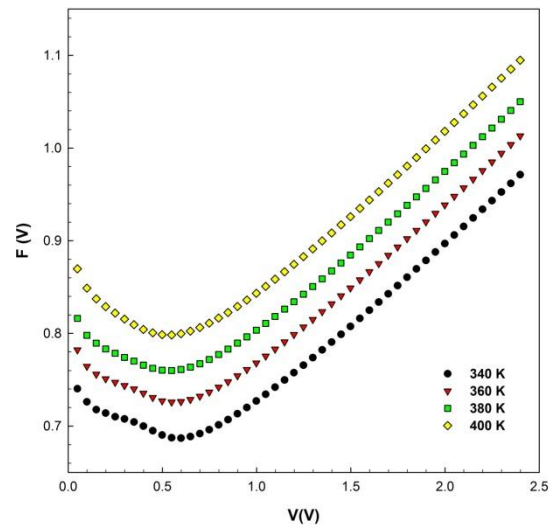


Fig. 3.  $F(V)$  vs  $V$  plot of the Au/TiO<sub>2</sub>/n-Si SBDs at various temperatures.

#### 4. Conclusion

The forward and reverse bias *I-V* characteristics of Au/TiO<sub>2</sub>/n-Si SBDs have been investigated in the temperature range of 340-400 K by steps of 20 K. The temperature dependence of electrical parameters were extracted from forward bias *I-V* measurement. Experimental results show that  $\Phi_b$  and  $R_s$  values are found to be changing depending on temperature. The values of  $\Phi_b$  and  $R_s$  have been obtained from two methods: Cheung's and Norde methods. It were found that  $\Phi_b$  and  $R_s$  values obtained from the forward bias ln(*I-V*) curves by applying Cheung's and Norde methods were in good agreement with each other. In summary, in the present study, TiO<sub>2</sub> thin films were deposited on polycrystalline n-type Silicon (Si) substrate using DC magnetron sputtering system. In order to improve the crystal quality, thermal annealing process were done at 700 °C. After then, For the electrical characterization, in the thermal evaporation system ohmic and rectifier contacts were formed using high purity Au. Furthermore,  $\Phi_b$  and  $R_s$  have been obtained from two methods and compare with each other.

#### Acknowledgement

This work is supported by the State of Planning Organization of Turkey under Grant no. 2011K120290.

#### Reference

- [1] L. Yang, S. S. Saavedra, N. R. Amstrong, J. Hayes, *Anal Chem.* **66**, 1254 (1994).
- [2] K. Zawreska, M. Radeska, M. Rekas, *Thin Solid Films* **310**, 161 (1997).
- [3] S. Ito, S. Kimatura, S. Yanagia, *Solar Energy Mater Solar Cell* **76**, 3 (2003).
- [4] H. Kostlin, G. Frank, G. Hebbinghaus, H. Auding, K. Denisen, *Journal of Non-Crystalline Solids* **218**, 347 (1997).
- [5] W. Yang, J. Marino, A. Monson, C.A. Woden, *Semiconductor Science and Technology* **21**, 1573 (2006).
- [6] N. R. Mathews, E. R. Morales, M. A. Cortes-Jacome, J. A. Toleda-Antonio, *Solar Energy* **83**, 1499 (2009).
- [7] L. Miao, P. Jin, K. Kaneko, A. Terai, N. Nabatova-Gabain, S. Tanemura, *Applied Surface Science*, **212-213**, 255 (2003).
- [8] M. K. Hudait, P. Venkateswarlu, S. B. Krupanidhi, *Solid State Electron* **45**, 133 (2001).
- [9] S. Chand, S. Bala, *Semicond. Sci. Technol* **19**, 82 (2004).
- [10] J. Osvald, *Solid-State Electron* **50**, 228 (2006).
- [11] K. Ejderha, N. Yıldırım, B. Abay, A. Turut, *J. Alloys Comp* **484**, 870 (2009).
- [12] V. Janardhamam, A. A. Kumar, V. R. Reddy, P. N. Reddy, *J. Alloys Comp* **485**, 467 (2009).
- [13] A. Tataroğlu, Ş. Altındal, *J. Alloys Comp.* **484**, 405 (2009).
- [14] P. Durmuş, Ş. Altındal, A. Tataroğlu, *J. Optoelectron. Adv. Mater.* **12**(7), 1472 (2010).
- [15] T. Tunç, İ. Dökme, Ş. Altındal, İ. Uslu, *Optoelectron. Adv. Mater. – Rapid Commun.* **4**(7), 947 (2010).
- [16] M. Soylu, F. Yakuphanoğlu, W. A. Farooq, *Optoelectron Adv. Mater. – Rapid Commun.* **5**(2), 135 (2011).
- [17] S. K. Cheung, N. W. Cheung, *Appl. Phys. Lett.* **49**, 85 (1986).
- [18] H. Norde, *J. Appl. Phys.* **50**, 5052 (1979).
- [19] R. T. Tung, *Mat. Sci. Eng. R* **35**, 1 (2001).
- [20] A. Gümüş, A. Türüt, N. Yalçın, *J. Appl. Phys.* **91**, 245 (2002).
- [21] E. H. Rhoderick, R. H. Williams, *Metal-Semiconductor Contacts*, second ed., Clarendon Press, Oxford. (1988).
- [22] K. E. Bohlin, *J. Appl. Phys.* **60**, 1223 (1986).

\*Corresponding author: kinacib@yahoo.com  
bariskinaci@gazi.edu.tr